

PHASE-FIELD MODELING OF CELLULAR AND DENDRITIC SOLIDIFICATION MICROSTRUCTURES



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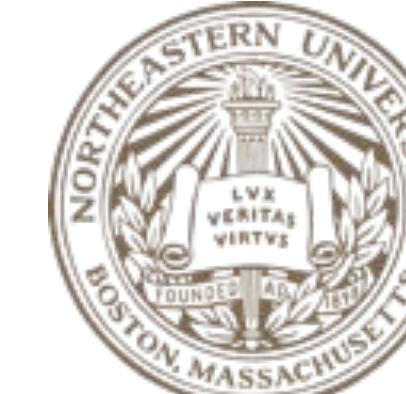
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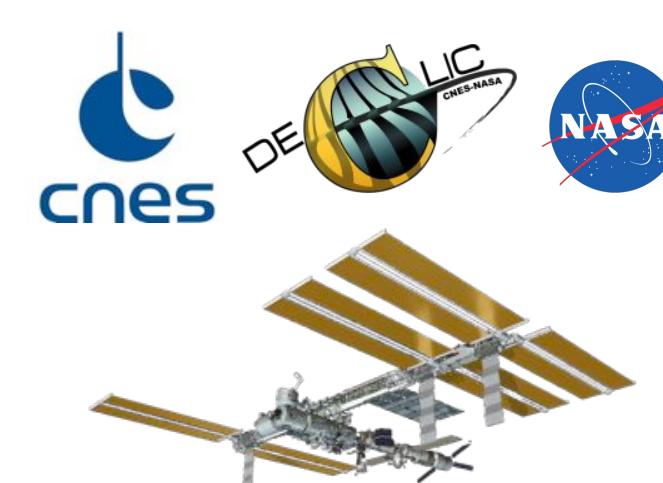


INTRODUCTION, CONTEXT, OBJECTIVE

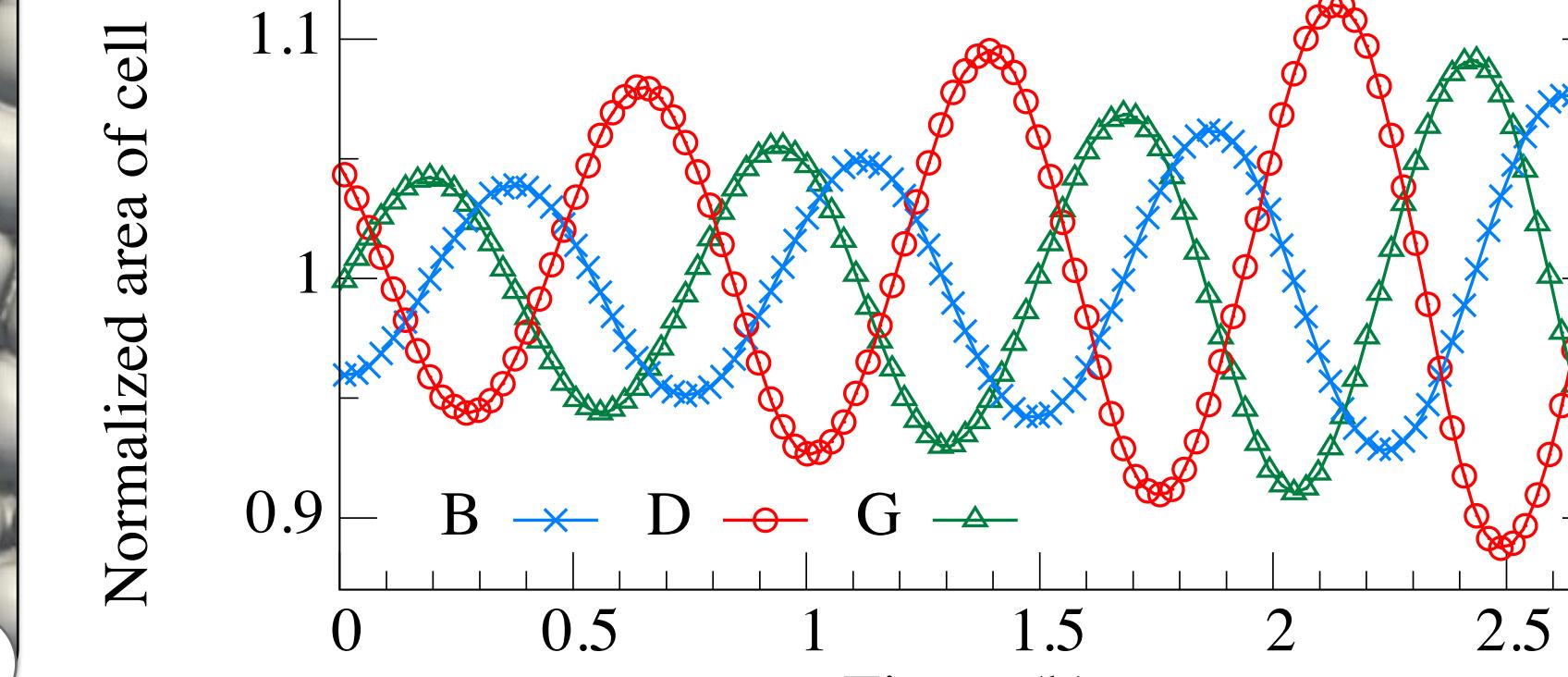
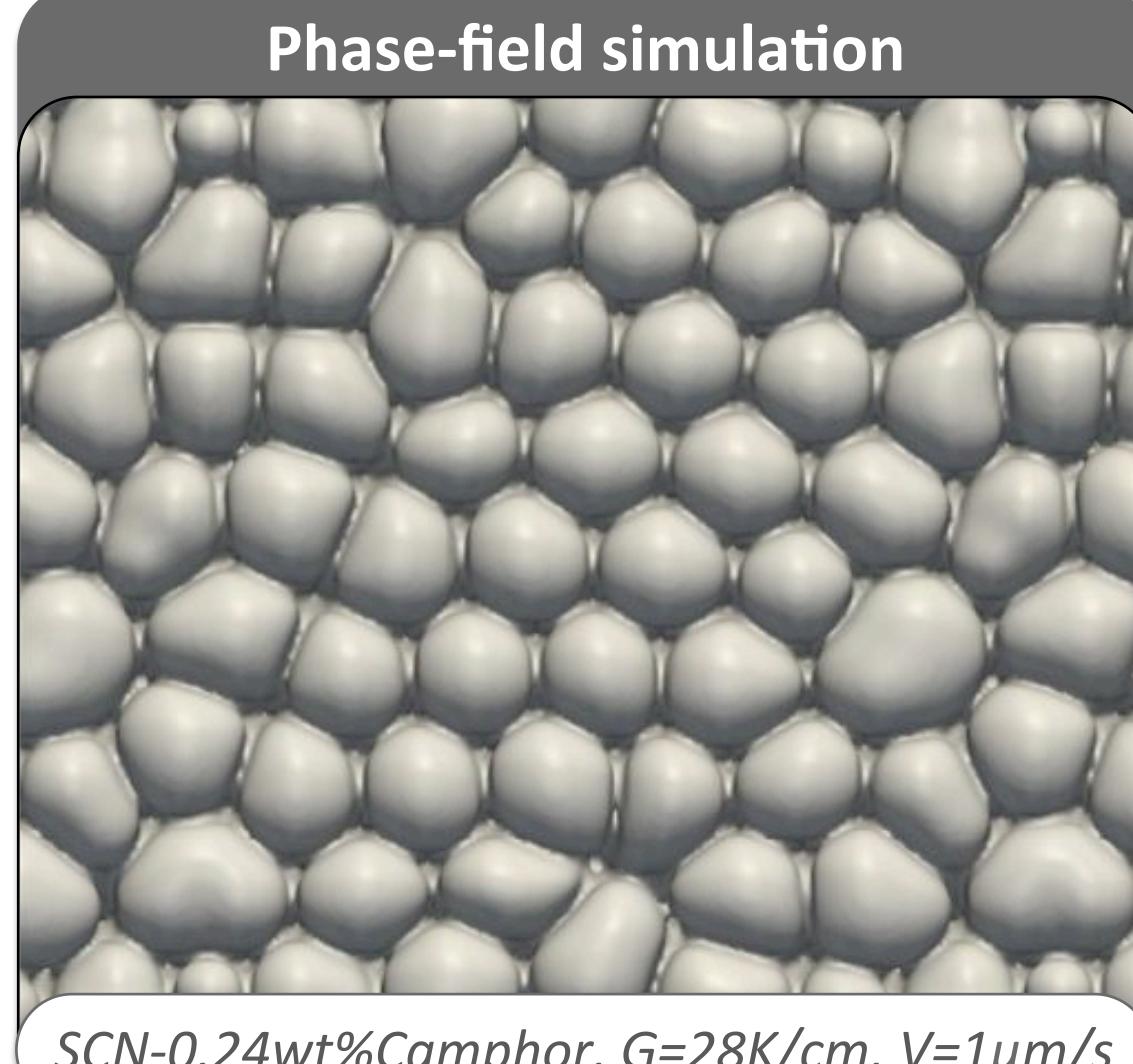
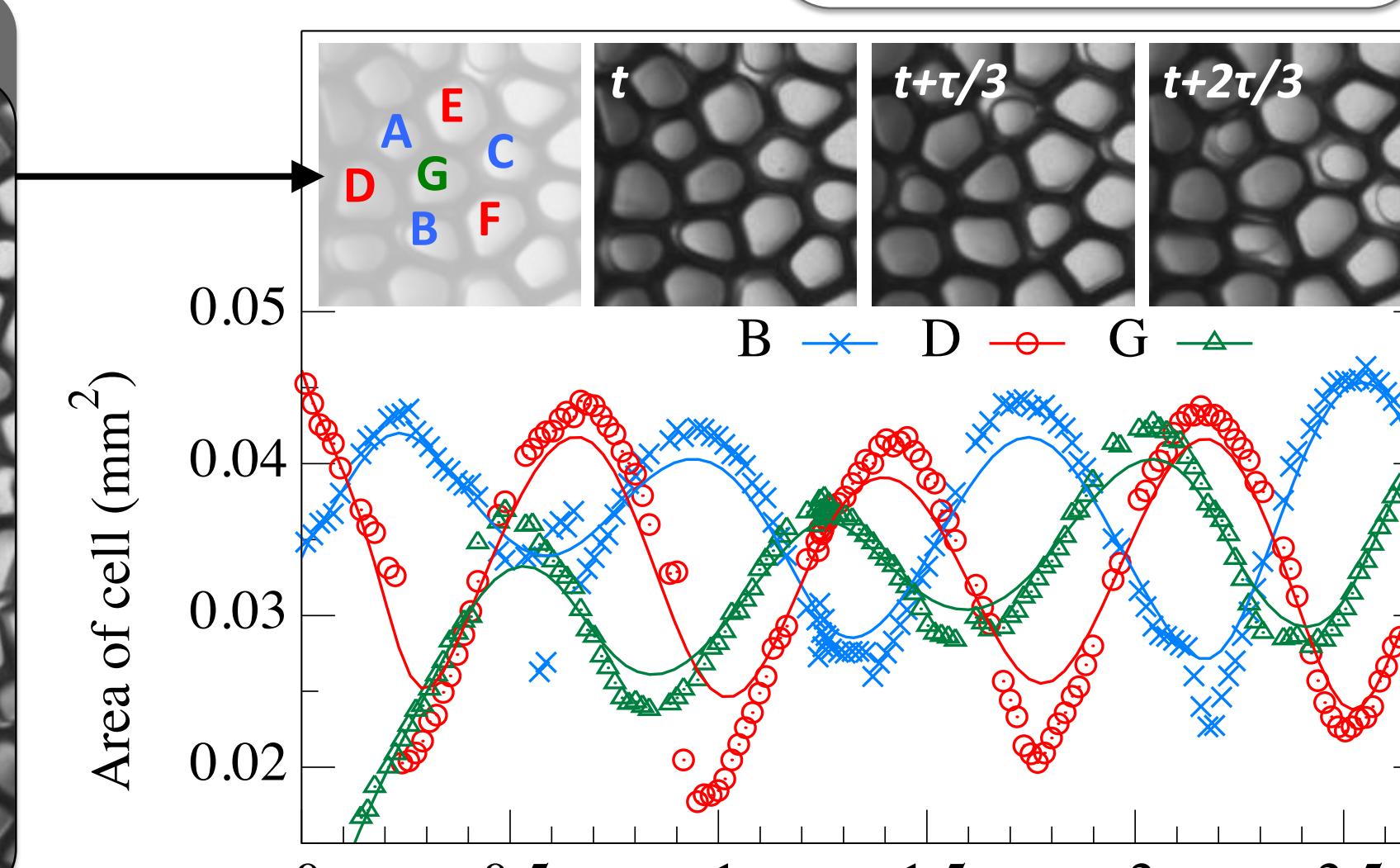
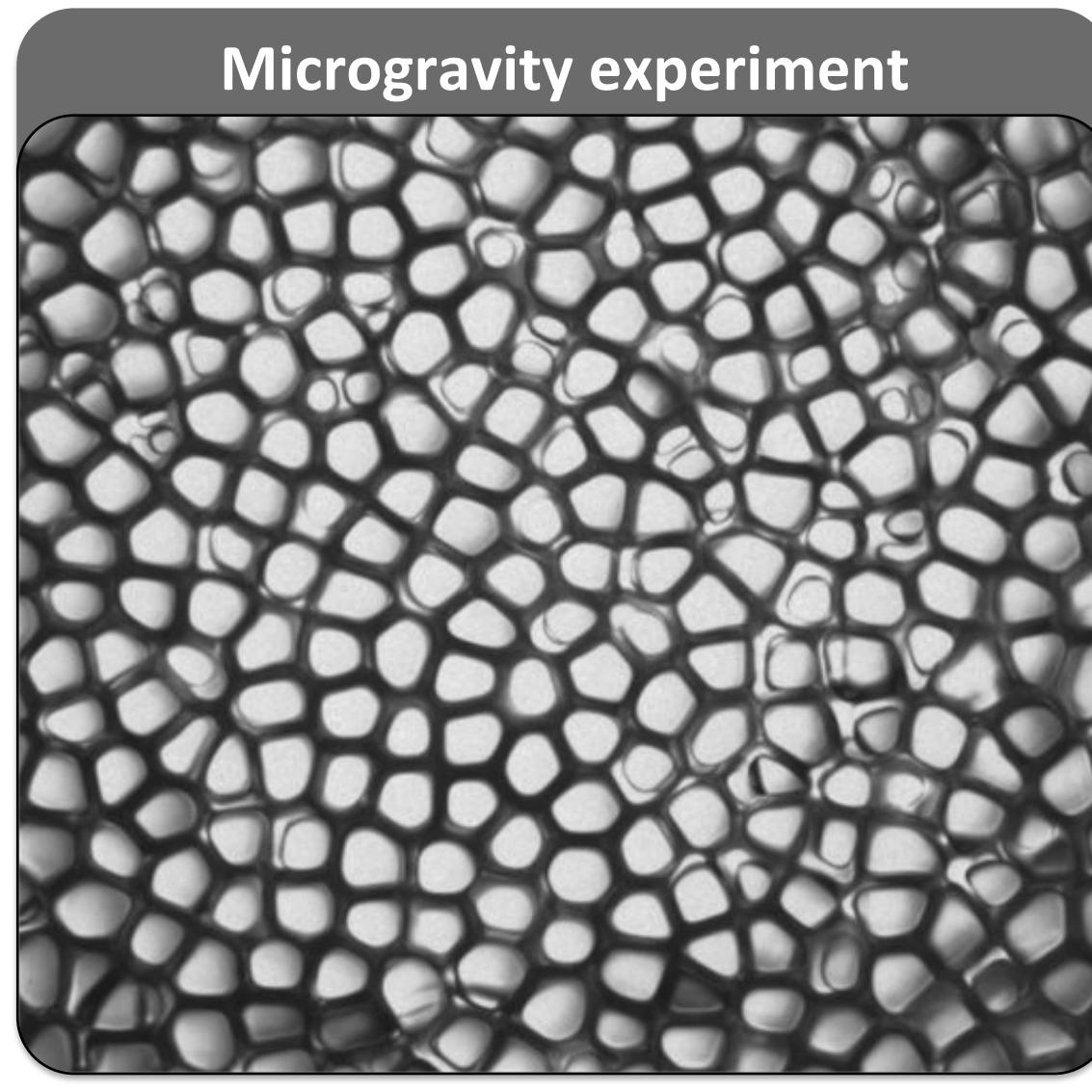
- ✓ Solidification microstructures crucial to metallurgical R&D
- ✓ Existing quantitative crystal growth phase-field (PF) model [1]
- ✓ Emerging highly-parallel calculation architectures (GPUs)
- ⇒ Objective: Implement a computationally efficient PF code of crystal growth capable of quantitative prediction, and use it extensively to study complex solidification scenarios
- Approach: Systematic quantitative comparisons with state-of-the-art well-controlled experiments and literature

CELLULAR OSCILLATORY GROWTH DYNAMICS [3]

- Device for the study of Critical Liquids and Crystallization (DECLIC)
 - Microgravity onboard the International Space Station
 - Directional Solidification Insert
- In situ observation of SCN-Camphor alloy
 - Oscillatory behavior of entire cellular arrays
 - Phase coherence only in ordered regions
 - ⇒ Localized breathing mode oscillations
 - Tip splitting instabilities prevent global order



Oscillation period
Experiment: $\tau \approx 46$ min
Simulation: $\tau \approx 48$ min



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PHASE-FIELD MODEL [1]

- Quantitative model for dilute binary alloy directional solidification [1]
- Graphic Processing Units (GPU) massively parallel implementation

Model equations

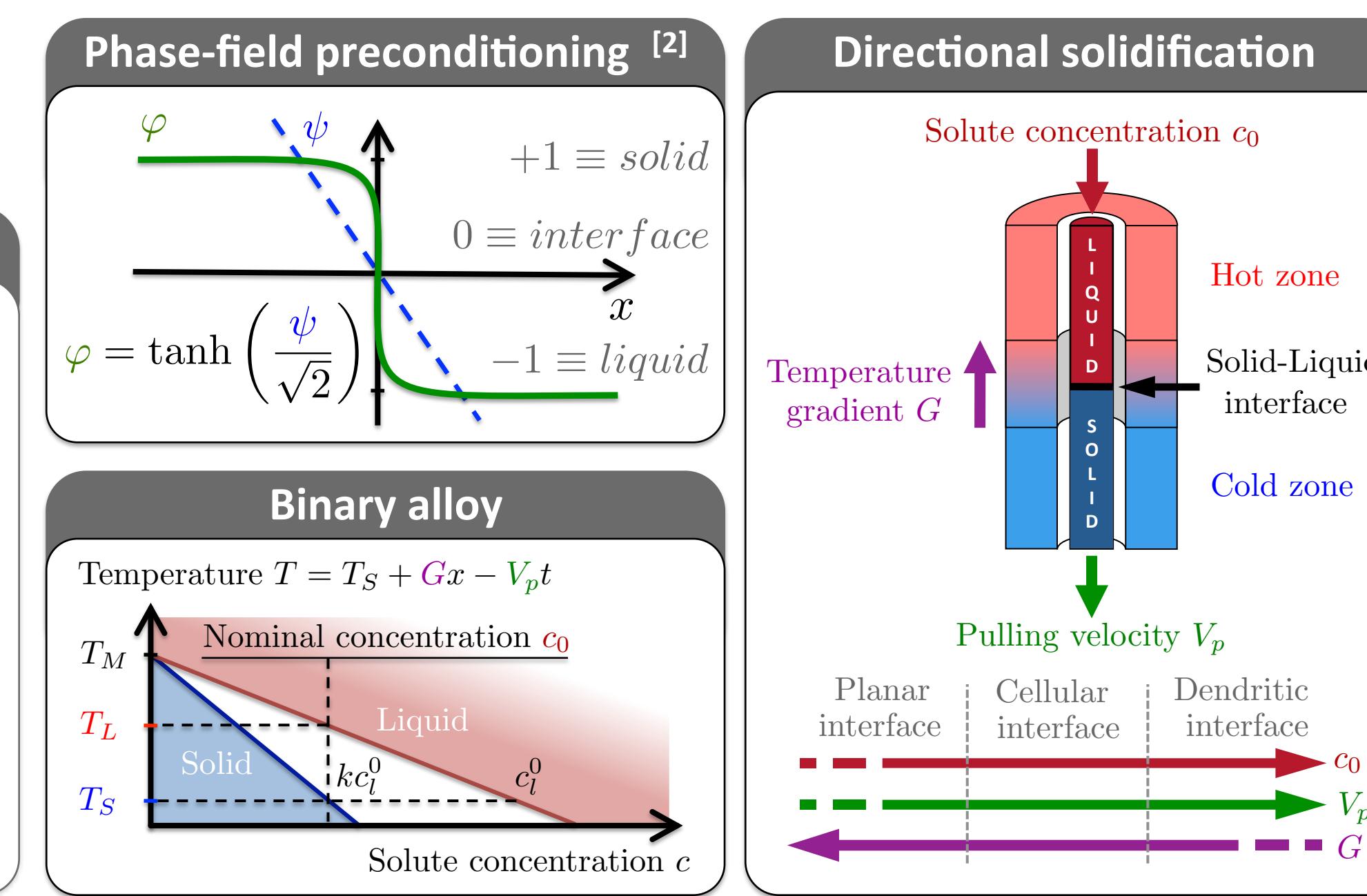
Preconditioned phase-field ψ

$$\left[1 - (1-k)\frac{x - \tilde{V}_p t}{\tilde{l}_T}\right] a_s(\theta)^2 \frac{\partial \psi}{\partial t} = \vec{\nabla} \cdot [a_s(\theta)^2 \vec{\nabla} \psi] + \sum_{m=x,y} \left[\partial_m \left(|\vec{\nabla} \psi|^2 a_s(\theta) \frac{\partial a_s(\theta)}{\partial (\partial_m \psi)} \right) \right] + \varphi \sqrt{2} \left(1 - a_s(\theta)^2 |\vec{\nabla} \psi|^2 \right) - \lambda (1 - \varphi^2) \sqrt{2} \left[\vec{U} + \frac{x - \tilde{V}_p t}{\tilde{l}_T} \right]$$

Solute supersaturation, \vec{U}

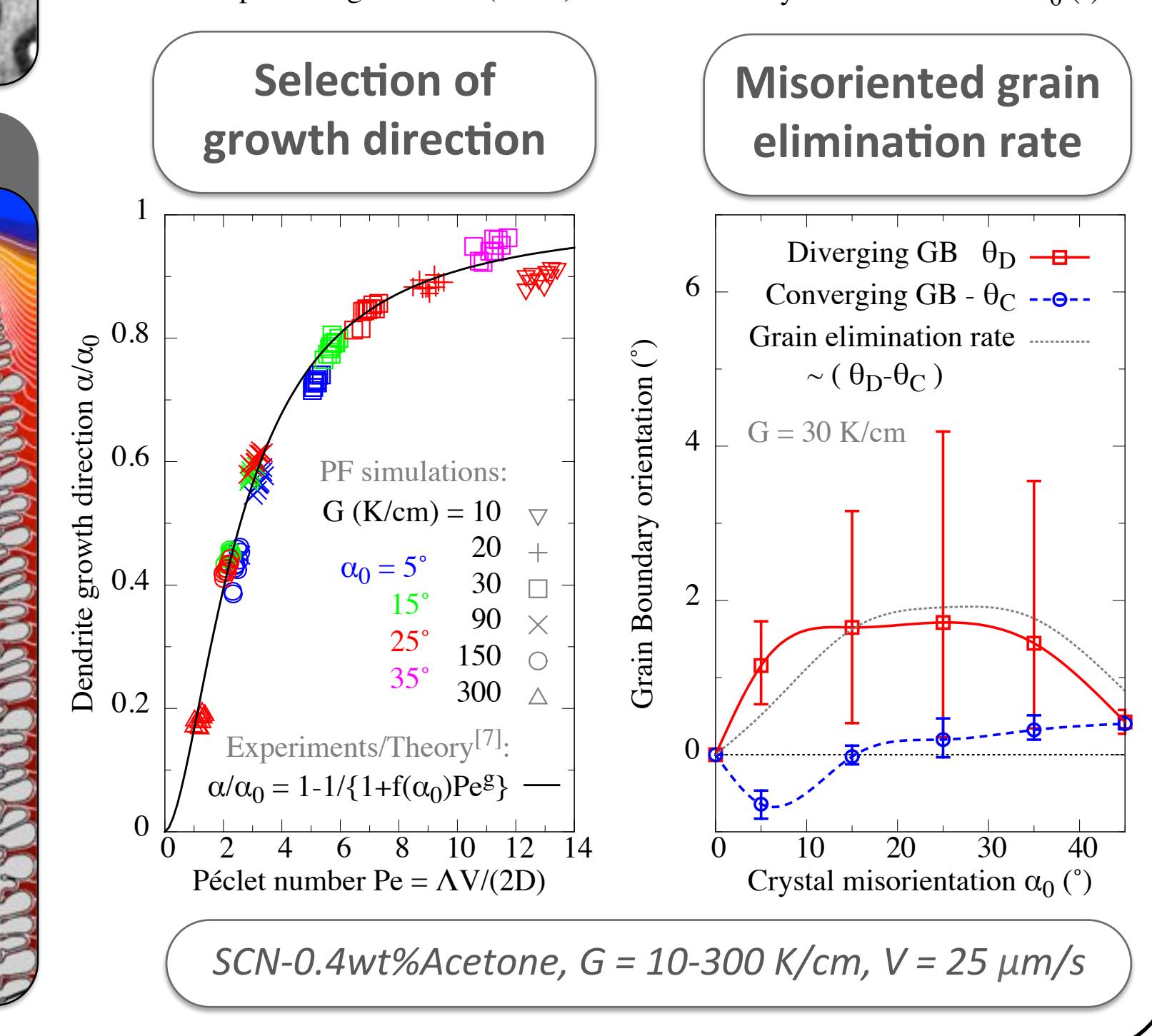
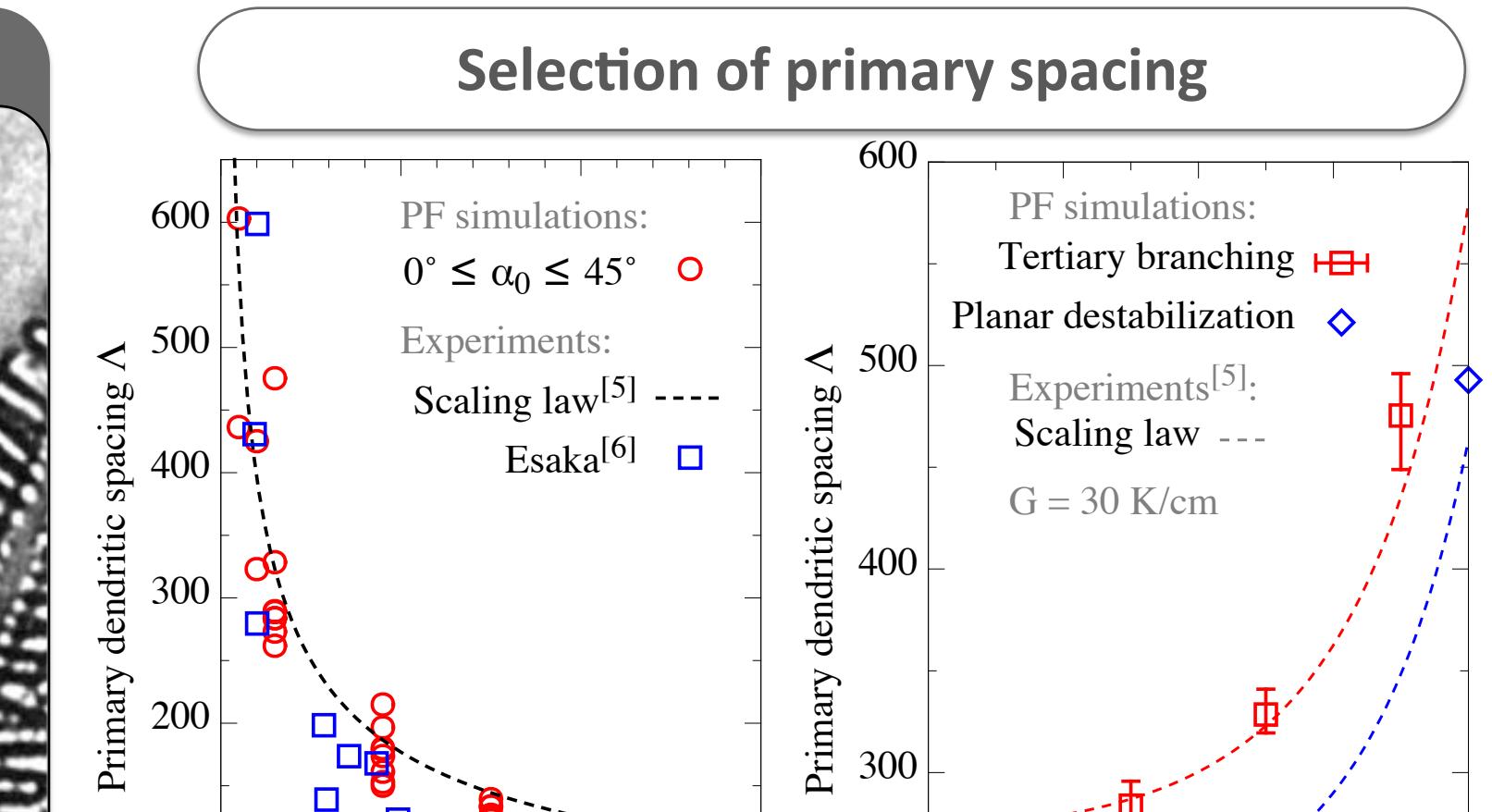
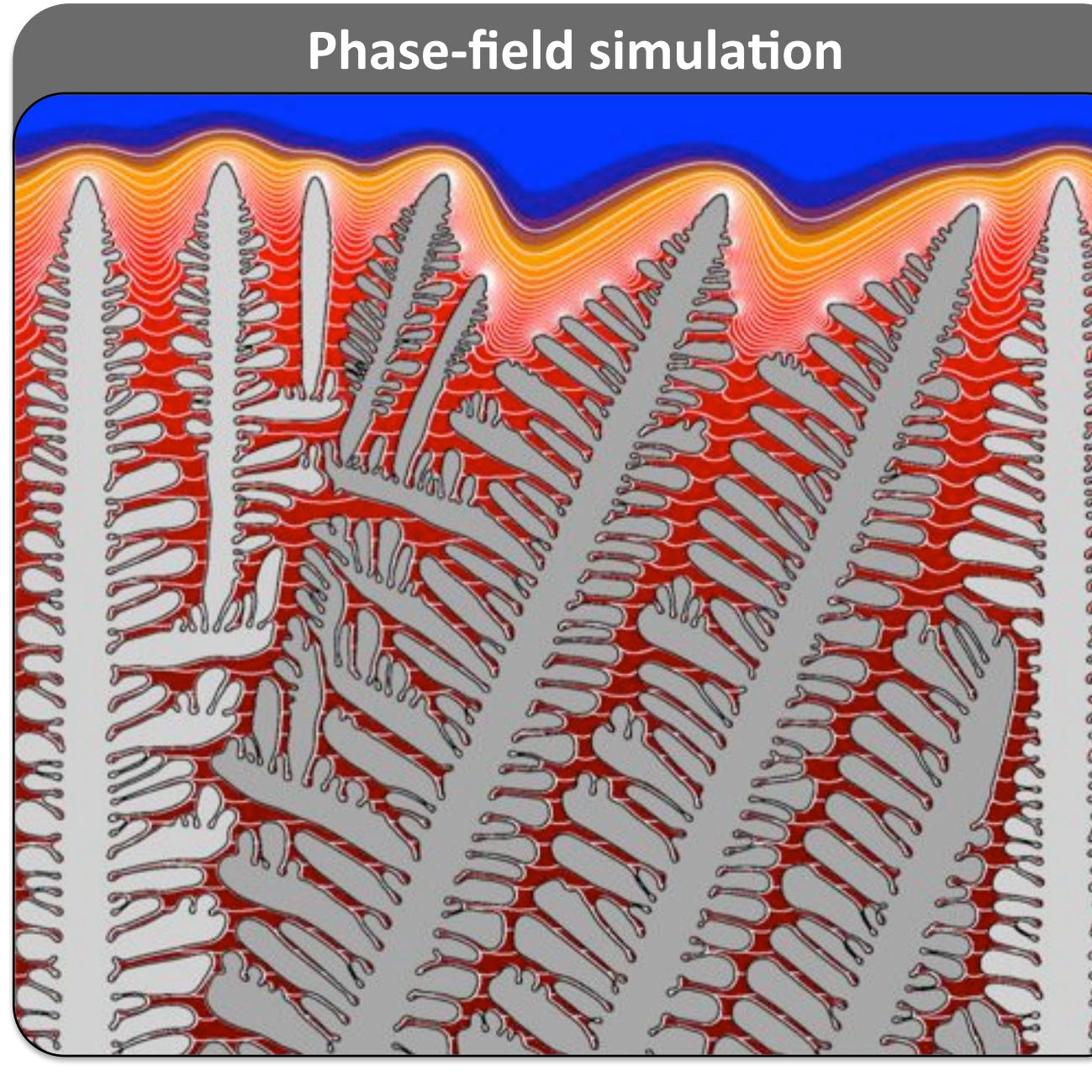
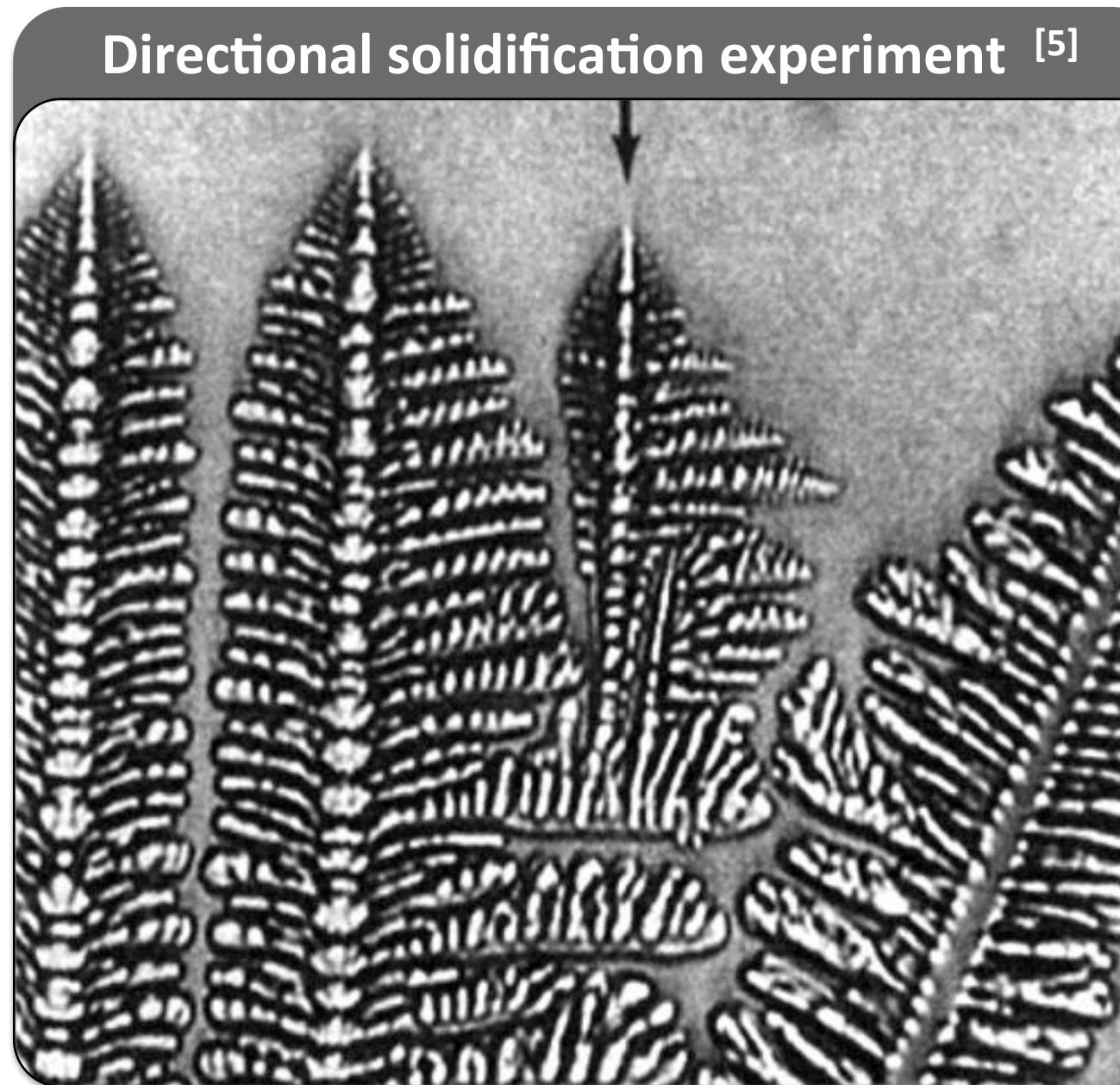
$$(1 + k - (1-k)\varphi) \frac{\partial \vec{U}}{\partial t} = \tilde{D} \vec{\nabla} \cdot [(1-\varphi) \vec{\nabla} \vec{U}] + \vec{\nabla} \cdot \left[\left(1 + (1-k)\vec{U} \right) \frac{(1-\varphi^2)}{2} \frac{\partial \psi}{\partial t} \frac{\vec{\nabla} \psi}{|\vec{\nabla} \psi|} \right] + \left[1 + (1-k)\vec{U} \right] \frac{(1-\varphi^2)}{\sqrt{2}} \frac{\partial \psi}{\partial t}$$

where the supersaturation derives from the solute concentration, c , as $\vec{U} = \frac{1}{1-k} \left[\frac{c/c_l^0}{(1-\varphi)/2 + k(1+\varphi)/2} - 1 \right]$ and c_l^0 , k , \tilde{D} , λ , \tilde{V}_p and \tilde{l}_T are alloy/process/computational parameters.



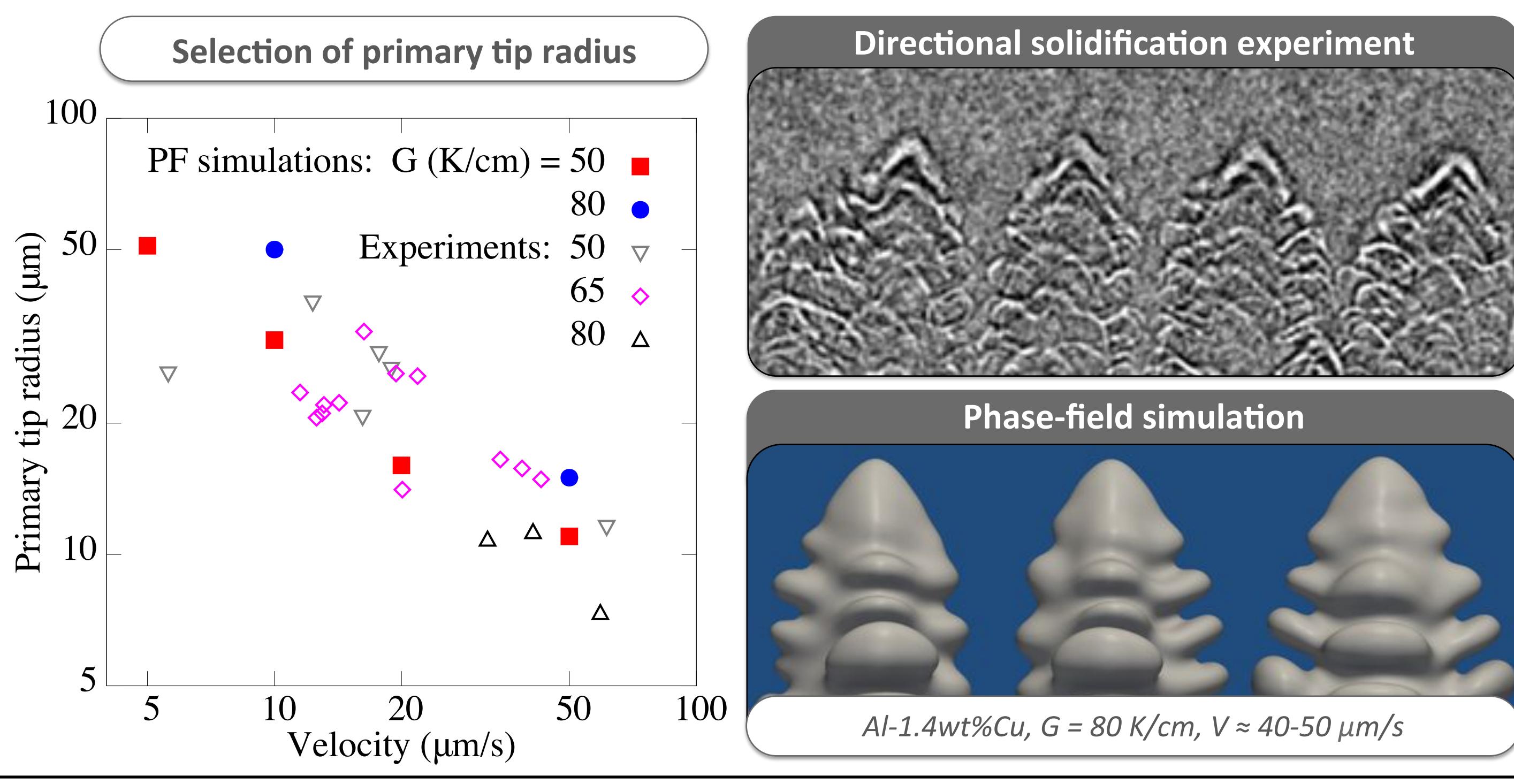
DENDRITIC GRAIN GROWTH COMPETITION [4]

- Computationally intensive study of a bi-crystal SCN-acetone alloy
 - Influence of the temperature gradient and crystal orientation
- Quantitative comparison to experimental values and scaling laws
 - Selection of primary spacing and dendrite growth direction
- Microscopic thermal fluctuations at the origin of sidebranching critically influences the macroscopic grain boundary orientations
- Grain elimination rate is not monotonic with crystal misorientation
- Spacing selected by branching > spacing after planar destabilization



DENDRITE TIP SELECTION IN METALLIC ALLOYS

- X-ray imaging of thin sample Al-Cu directional solidification
 - Measure of dendritic tip radius at different growth velocities



SUMMARY, CONCLUSIONS, OUTLOOK

- ✓ Phase-Field model parallel implementation on GPUs
 - ⇒ Quantitative predictions at the scale of experiments
- Cellular oscillations in 3D more complex than quasi-2D experiments: Long range coherence linked to spatial order
- Stochastic selection of dendritic GB orientation in polycrystals
- Outlook:
 - Dendritic regime in microgravity experiments
 - Dendritic growth competition in 3D
 - Effect of convection on microstructure selection

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